

thelecture thenextlecture

2.4.1 Laue Method

1912: Max von Laue (assisted by Paul Knipping and Walter Friedrich). CuSO₄ and ZnS. Broad x-ray spectrum – single crystal

SOLID STATE PHYSICS Lecture 5

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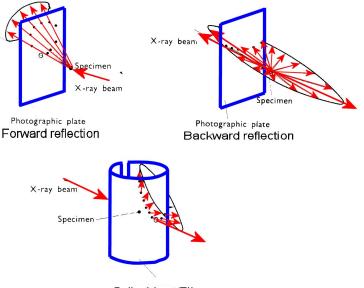
Structure & Diffraction

Crystal Diffraction (continued)

2.4 Experimental Methods

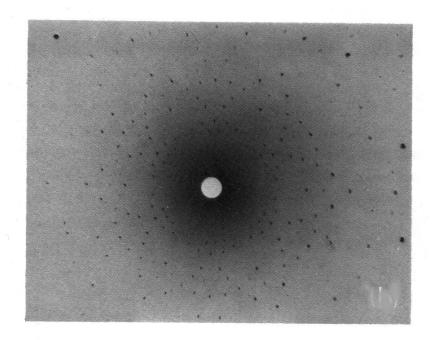
Notes:

- examples show *photographic film*, for x-rays.
- Can also use electronic detection for x-rays.
- \bullet Need counters (e.g. BF_3) for neutrons.
- Information:
 - Positions of lines (geometry)
 - Intensities of lines (electronics, or photogrammetry to measure darkness of lines on films)



Cylindrical Film

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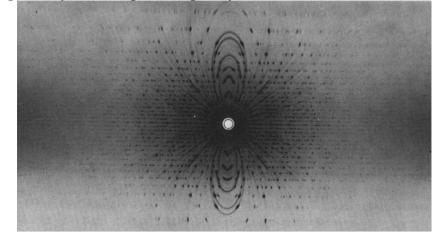
Forward scattering Laue image of hexagonal crystal.

Shows crystal symmetry – when crystal appropriately oriented. Use for aligning crystal for other methods.

Range of λ , so cannot determine a from photographic image, but if outgoing wavelengths can be measured, *can* use to find lattice parameters.

2.4.2 Rotating Crystal Method

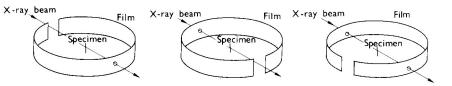
Single x-ray wavelength – single crystal rotated in beam.



Either full 360° rotation (as above) or small (5 to 15°) oscillations.

2.4.3 Powder Methods

Single x-ray wavelength – finely powdered sample. Effect similar to rotating crystal, but rotated about all possible axes.





X-ray powder diffraction pattern of ${\rm NaClO}_3$ taken with ${\bf C}{\bf u}K\alpha$ radiation.



X-ray powder diffraction pattern of ${\rm SiO}_2$ taken with $\mbox{Cu}K\alpha$ radiation.

Powder diffraction patterns are often used for identifying materials.

So if a plane wave with wavevector \mathbf{k}_f is scattered from the crystal, it is the sum of the waves scattered by all the atoms, or

Total Wave =
$$S A \exp[i(\mathbf{k}_f \cdot \mathbf{r} - \omega t)] \sum_{I} \exp[i(\mathbf{k}_i - \mathbf{k}_f) \cdot \mathbf{r}_I].$$

Write $\Delta k = \mathbf{k}_f - \mathbf{k}_i$:

Total Wave =
$$S A \exp[i(\mathbf{k}_f \cdot \mathbf{r} - \omega t)] \sum_{I} \exp[-i\Delta \mathbf{k} \cdot \mathbf{r}_I],$$

and as the amplitude of the outgoing wave $\exp[i(\mathbf{k}_f.\mathbf{r}-\omega t)]$ is 1,

Total Amplitude
$$\propto S \sum_{I} \exp[-i\Delta \mathbf{k} \cdot \mathbf{r}_{I}].$$
 (1)

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2.5 Mathematics of Diffraction

2.5.1 Monatomic Structure

Incoming plane wave

$$\psi_i = A \exp[i(\mathbf{k}_i \cdot \mathbf{r} - \omega t)]$$

Scattered by the atom in unit cell I at r_I .

Assume scattered amplitude is S A – all the unit cells are the same, so independent of I.

When incident wave hits atom, it is

$$A \exp[i(\mathbf{k}_i \cdot \mathbf{r}_I - \omega t)].$$

It is scattered with a different wave-vector, \mathbf{k}_f , so from the atom to a point r its phase changes by $\mathbf{k}_f \cdot (\mathbf{r} - \mathbf{r}_I)$. The scattered wave is thus

$$S A \exp[i(\mathbf{k}_i \cdot \mathbf{r}_I - \omega t)] \exp[i\mathbf{k}_f \cdot (\mathbf{r} - \mathbf{r}_I)]$$

or

$$S A \exp[i(\mathbf{k}_f \cdot \mathbf{r} - \omega t)] \exp[i(\mathbf{k}_i - \mathbf{k}_f) \cdot \mathbf{r}_I]$$

2.5.2 The Reciprocal Lattice

Define a new set of vectors $({\bf A},{\bf B},{\bf C})$ with which to define $\Delta {\bf k}.$ Require

a.**A** =
$$2\pi$$
 , **a**.**B** = 0 , **a**.**C** = 0
b.**A** = 0 , **b**.**B** = 2π , **b**.**C** = 0
c.**A** = 0 , **c**.**B** = 0 , **c**.**C** = 2π
(2)

In general,

$$\mathbf{A} = \frac{2\pi \mathbf{b} \times \mathbf{c}}{\mathbf{a}.\mathbf{b} \times \mathbf{c}}$$
$$\mathbf{B} = \frac{2\pi \mathbf{c} \times \mathbf{a}}{\mathbf{a}.\mathbf{b} \times \mathbf{c}}$$
$$\mathbf{C} = \frac{2\pi \mathbf{a} \times \mathbf{b}}{\mathbf{a}.\mathbf{b} \times \mathbf{c}}$$
(3)

The vectors (A, B, C) define the *reciprocal lattice*. For simple cubic system, reciprocal lattice vectors are just $2\pi/a$ along the x, y and z axes.

Lattice	Reciprocal Lattice
Simple cubic	Simple cubic
FCC	BCC
BCC	FCC
Hexagonal	Hexagonal

So we see

- we have a strong reflection when $\Delta \mathbf{k}$ is a reciprocal lattice vector;
- remembering that $\Delta \mathbf{k}$ is perpendicular to the reflecting plane, an (hkl) reflection has $\Delta \mathbf{k} = h\mathbf{A} + k\mathbf{B} + l\mathbf{C}$.

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2.5.3 The Scattered Amplitude

Let

$$\Delta \mathbf{k} = h\mathbf{A} + k\mathbf{B} + l\mathbf{C},$$

and remember that our structure is periodic:

$$\mathbf{r}_I = n_1 \mathbf{a} + n_2 \mathbf{b} + n_3 \mathbf{c}$$

Immediately we have

$$\Delta \mathbf{k} \cdot \mathbf{r}_I = 2\pi (hn_1 + kn_2 + ln_3).$$

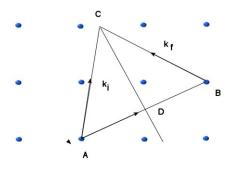
So

$$\sum_{I} \exp[-i\Delta \mathbf{k} \cdot \mathbf{r}_{I}] = \sum_{n_{1}} \sum_{n_{2}} \sum_{n_{3}} \exp[-2\pi i(hn_{1} + kn_{2} + ln_{3})]$$
$$= \left\{ \sum_{n_{1}} e^{-2\pi i hn_{1}} \right\} \left\{ \sum_{n_{2}} e^{-2\pi i kn_{2}} \right\} \left\{ \sum_{n_{3}} e^{-2\pi i ln_{3}} \right\}$$

Sums, in principle, go over $-\infty < n_i < \infty$, or at least over a very large range $1 \le n_i \le N_i$.

Phases lead to cancellation unless h, k and l are integers, when each term is 1 and total amplitude is $SN_1N_2N_3$.

2.6 The Laue Construction



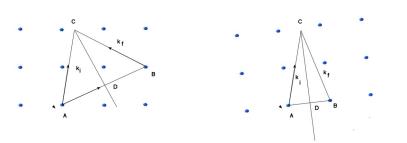
This is a diagram in the *reciprocal lattice*.

Just as the lattice is an abstract mathematical object, so is the reciprocal lattice.

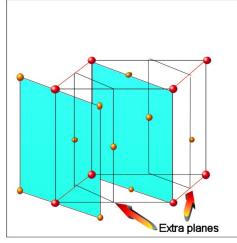
Neither \mathbf{k}_i nor \mathbf{k}_f need to be reciprocal lattice vectors, but $\mathbf{k}_f - \mathbf{k}_i$ is.

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Note that only certain special incident directions of \mathbf{k}_i will give a diffracted signal.



but face-centred cubic has extra atoms in the orginal planes and between them:



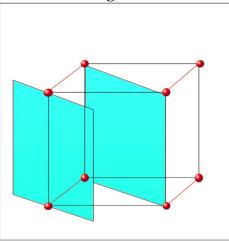
These extra planes have the same number of atoms as the original (110) planes. But if the original planes correspond to a path length difference of λ , these have path length difference of $\lambda/2$ – their signals will be *out of phase*. If the atoms are all the same, the (110) reflection will be missing. If the atoms are different, the amplitude of the (110)

reflection will be reduced.

2.7 Non-Monatomic Structures

2.7.1 Simple Treatment

Example: an FCC structure (thought of as simple cubic with a basis of two atoms, one at (0, 0, 0), three more at $(\frac{1}{2}, \frac{1}{2}, 0)$, $(\frac{1}{2}, 0, \frac{1}{2})$, $0, \frac{1}{2}, \frac{1}{2})$. For simple cubic, there is a strong reflection from (110) planes:



These *missing orders* tell us something about the structures:

- simple cubic no missing orders;
- \bullet fcc only see (hkl) where h,k and l are all even OR all odd.
- **bcc only see** (hkl) where h + k + l is even.

Summary

- Experimental methods broad-band or single-wavelength;
- Bragg's law explained by von Laue's treatment;
- Scattering treatment;
- The reciprocal lattice;
- Effect of atomic basis.

Next:

- Detailed treatment of structure with a basis;
- Other information from diffraction;
- Binding of crystals.